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1 **Cellulose nanofibrils and silver nanowires active coatings for the development**
2 **of antibacterial packaging surfaces**

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18 **Abstract:** An active ink composed of cellulose nanofibrils and silver nanowires was deposited on flexible
19 and transparent polymer films using the bar coating process, achieving controlled thicknesses ranging
20 from 200 nm up to 2 µm. For 350 nm thick coating on polyethylene terephthalate films, high transparency
21 (75.6% transmittance) and strong reduction of bacterial growth equal to 89.3% and 100% was noted
22 respectively against Gram-negative *Escherichia Coli* and Gram-positive *Staphylococcus Aureus* bacteria
23 using AATCC contact active standard test. Retained antibacterial activity was found with films produced
24 by reverse gravure roll-to-roll process, showing the promising capability of this antibacterial solution to be
25 deployed industrially. Finally, the same ink was also deposited on polylactic acid substrate to investigate
26 barrier properties: for 350 nm thick coating, a reduction of 49% of oxygen transmission rate (dry
27 conditions) and 47% reduction of water vapor transmission rate was noted, proving the enhanced barrier
28 properties of the coatings.

29 **Keywords:** cellulose nanofibrils, silver nanowires, transparent coatings, antibacterial activity, barrier
30 properties, up-scaling

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31 **1. Introduction**

32 In a high consumption society, production and global movement of food products is considerable
33 and increasing continuously (Food and Agriculture Organization of the United Nations, 2019). Food
34 products are exposed to harsh conditions throughout all of the supply chain stages and packaging is today
35 expected to meet more and more requirements. Recent innovation has focused on the development of
36 active packaging, which refers to the ability of a package to respond to an external stimulus (Yildirim et
37 al., 2018). Within this framework, active packaging aims at increasing the shelf-life of a product by
38 optimizing the condition inside the package, thus improving food quality and reducing food waste.

39 Antimicrobial food packaging has been studied extensively over the past 20 years and aspires to
40 reduce pathogenic contamination against bacteria, fungus or viruses (Sofi et al., 2018) with a preferred
41 mode of action working by contact for consumer safety and environmental concern. (Kaur & Liu, 2016).
42 Silver nanoparticles (Ag NPs) have recently attracted a lot of attention since they display strong
43 antibacterial activity and can help fighting against the increasing antibiotic resistance in pathogens (Rai et
44 al., 2012). It has been established that the antibacterial effect of Ag NPs is attributed to the nanoscale of
45 the particles itself along with the continuous release of silver ions (Morones et al., 2005). The morphology
46 of Ag NPs plays a major role on their antimicrobial activity: the smaller the nanoparticle (the higher
47 surface area) the more active they are against bacteria (Carlson et al., 2008). The shape of the nanoparticle
48 is also important and the antibacterial activity decreases on progression from pyramids, cubes, spheres, to
49 wire-like shape (Hong et al., 2016; Pal et al., 2007). Different parameters also influence the antibacterial
50 activity of Ag NPs such as surface charge, the presence of residual synthesis surfactants or capping agent,
51 and Ag NPs crystallinity. (El Badawy et al., 2011; Kvítek et al., 2008; Smetana et al., 2008).

52 One way to use Ag NPs in packaging applications is to incorporate them into a nanocomposite to limit
53 nanoparticle migration. Indeed, Ag NPs have been recognized as potentially hazardous for health as
54 studies have focused on evaluating their toxicity against in vivo and in vitro studies. There is still a lack of
55 information and knowledge concerning the global health hazard on human, but Ag NPs are suggested to
56 be mainly dangerous by oral and inhalation adsorption into the human body and silver accumulation in
57 different organs leading to specific malfunctions (Korani et al., 2015). The potent activity of Ag NPs is
58 dependent on a lot of factors such as size, surface chemistry, shape, etc. and more research is needed to
59 have a full overview of the toxicity of Ag NPs toward the human body. It then appears obvious that the
60 leaching of Ag NPs in antibacterial composite for food packaging application is important parameter to
61 address. Nanocellulose materials are biobased, biodegradable and biocompatible nanomaterials that are
62 defined as cellulose particles with a nano-scale size for at least one of their dimensions. One of the
63 established class of nanocellulose are cellulose nanofibrils (CNF) which come from the mechanical

64 fibrillation of pulp suspension (Herrick et al., 1983; Turbak et al., 1983). CNF have been of particular
65 interest because they display interesting properties for packaging applications: high oxygen barrier
66 property, vast chemical functionalization possibilities and good mechanical properties. CNF can be used
67 either alone as a free standing film or as a coating on classic plastic packaging or paper substrates for its
68 reinforcement properties (Afra et al., 2016; Fukuzumi et al., 2009; Rodionova et al., 2012) and also
69 display good matrix properties that can be used successfully to immobilize nanoparticles.

70 Combinations of Ag NPs and nanocellulose composites have been investigated extensively by the
71 scientific community (Xu et al., 2017) and different composite preparation techniques exist. In the most
72 common in-situ synthesis methods, CNF serve as a template matrix for adsorption of silver salts, followed
73 by in-situ chemical reduction (Uddin et al., 2017; Yu et al., 2019). Covalent bonding is another option ,yet
74 to our knowledge, the only reported example comes from Ramaraju, Imae, & Destaye (2015), who
75 reported the immobilization of Ag NPs stabilized with a dendrimer by covalently bonding the NH₂-
76 terminated dendrimer on TEMPO-oxidized CNF. Finally, the physical mixing approach is just mixing
77 separately produced components. To our knowledge, the only example of simple mixing of CNF materials
78 with Ag NPs for antibacterial applications has been reported by Martins et al. (2012) who prepared
79 CNF/Ag NPs composites based on a polyelectrostatic assembly for antibacterial coating on a paper
80 substrate. Simple mixing is a straightforward easily up-scalable procedure, however immobilization of the
81 Ag NPs on cellulose materials is relatively poor (Ilić et al., 2009), leading to reduced life-time of the
82 system and possible non-desired release of Ag NPs.

83 This work presents the development of antibacterial packaging based on a combination of silver
84 nanoparticles and nanocellulose materials. Indeed, based on the precedent literature review, it was then
85 hypothesized that combining silver nanowires with cellulose nanofibrils could achieve easy-to-process
86 systems that can be deposited on flexible substrate to achieve contact killing antibacterial mode of action
87 along with improved barrier properties. In this study, the technical challenges were firstly to
88 straightforwardly produce a transparent and antibacterial coating on plastic polymer sheets. Secondly the
89 objectives were to prepare similar active coatings on biobased polymer sheets and to evaluate the
90 enhancement of their barrier property. Finally, the challenge of preparing active surfaces with a roll-to-roll
91 process for up-scaling purposes was addressed.

92 To tackle the three core challenges, a hybrid ink composed of TEMPO-oxidised cellulose nanofibrils and
93 silver nanowires was firstly deposited on polyethylene terephthalate substrates using laboratory-scale bar
94 coating process. Thickness and transparency of the coatings were investigated and they were tested against
95 bacterial contamination. The same technique was used to deposit the active ink on bio-based polylactic
96 acid sheets and the barrier properties of the produced films were characterized. Finally, a new lower cost

97 active ink was specifically formulated by the authors to assess the up-scaling possibility by using a reverse
98 gravure roll-to-roll coating process.

99 **2. Materials and methods**

100 **2.1. Materials**

101 A polyethylene terephthalate (PET) sheet substrate (Melinex® ST726 – 175 µm) and polylactic acid
102 (PLA) substrate (Earthfirst® BCWC – 75 µm) was respectively purchased from Dupont (France) and
103 Sidaplast (Belgium). A hybrid ink system composed of silver nanowires (Ag NWs) and TEMPO oxidised-
104 cellulose nanofibrils (T-CNF) in water (PolyBioWire® 9830C) was supplied by Poly-Ink (France) with an
105 approximate 1% wt total mass and 1:1 ratio given by the supplier. However to investigate a roll-to-roll
106 semi-industrial deposition process, a new specific formulation was prepared, using T-CNF suspension
107 (CNF-A13) from Betulium (Finland) with approximatively length of 50-400 nm and diameter of 5-15 nm,
108 at 5 %wt (charge density of $1.6 \pm 0.1 \text{ mmol.g}^{-1}$, estimated aspect ratio of 22.5) and using 2.5% wt Ag NWs
109 aqueous suspension (NGAP NF Ag-3170) from NanoGap (USA) with estimated length 10-50 µm,
110 diameter 50-100 nm (and so with an estimated aspect ratio of 400).

111 Escherichia Coli ATCC 8739 and Staphylococcus Aureus ATCC 6538 were purchased from
112 Microbiologics (USA). PCA (Plate Counting Agar) was purchased from BD Difco (USA) and contains
113 beef extract (3g.l^{-1}), peptone (5g.l^{-1}), agar (15g.l^{-1}). Standard nutrient broth 1 was purchased from Roth
114 (Germany) and contains beef extract (3g.l^{-1}), peptone (15g.l^{-1}), sodium chloride (6 g.l^{-1}) and glucose (1g.l^{-1}).
115 Sodium thiosulphate (>99%), L-histidine (>98.5%), potassium dihydrogen phosphate (>99%),
116 Tween®80 and calcium chloride (>96%) were also obtained from Roth (Germany). L α -phosphatidyl
117 choline (>99%) was purchased from Sigma Aldrich (France). All materials were used as received.

118 **2.2. Active layer production by bar coating process**

119 Coated films were first prepared using the supplied T-CNF/Ag NWs ink and bar coating process
120 (KCC101, Erichsen, Germany). Prior to coating, the films were corona treated (Klwar Calvatron SG2,
121 Germany): 2 passes, 90° rotation between the pass. The coatings were then oven-dried and different
122 thickness were achieved using different threaded rod diameter (Supplementary data). Coatings, corona
123 treatment and drying step parameters were optimized through extensive research with visual observation
124 of possible defects, transparency and sheet resistance measurements. Optimized corona treatment settings
125 are the following: 2.9 m.min^{-1} speed and 330 mA intensity for PET substrate, 3.5 m.min^{-1} speed and 150
126 mA intensity for PLA substrate. Optimized coatings and drying parameters are the following: coating
127 speed of 5.4 m.min^{-1} for both PET and PLA coatings, oven-drying step at 120°C for 60s for PET and at

128 room temperature overnight for PLA coatings. Each coating was performed at least three times to assess
129 consistency and reproducibility of the process.

130 **2.3. Structural and quality characterization**

131 Field-Emission Gun Scanning Electron Microscope (FEG-SEM) was used to investigate the hybrid
132 nanostructure after 2 nm-coating of Gold /Palladium, at 5.4 mm working distance and 3 kV accelerating
133 voltage. A cross-section through the coated samples was prepared using a LEICA UC6 (Germany)
134 ultramicrotome apparatus equipped with a diamond knife, at room temperature and with a cutting speed of
135 1 mm.s⁻¹. The thickness of the coatings was then estimated by image analysis using FIJI software
136 (Schindelin et al., 2012; Schneider et al., 2012). At least 6 measurements in three different positions in the
137 films were considered.

138 Transmittance (%) was measured on a UV-spectrophotometer (Shimadzu Manufacturing Inc., USA),
139 using the photometric mode. Six different measurements were conducted on each sample at 550 nm
140 wavelength. Sheet resistance (Ω /sq) of the coated layer was measured by a four-probe system (Jandel
141 Universal, USA). The measurement was repeated at six positions on the film and the average computed
142 along with associated standard deviation.

143 **2.4. Antibacterial characterization**

144 Prior to antibacterial testing, a leaching assay was performed by putting a 2 x 2 cm specimen of coating
145 (24 μ m estimated wet thickness) in 10 ml of deionized water (DI), under agitation (100 rotation per
146 minute) at 37°C. After 72 hours, the liquid media was recovered and full UV-vis spectra (300-700 nm
147 wavelength, 1 nm scan rate) were recorded on a UV-spectrophotometer (Shimadzu Manufacturing Inc.,
148 USA), and compared with Ag NWs suspensions at 0.005% wt.

149 All the glassware, consumables, tools and solutions used for the antibacterial characterization were
150 sterilized prior to use in autoclave for 20min at 120°C and 1.034 bar.

151 Antibacterial activity was assessed initially using an inhibition zone qualitative test following a modified
152 AFNOR EN 1104 standard (AFNOR standard, 2005). To exaggerate the antibacterial activity of the
153 samples, the test was carried out on multilayer coating (10 layers) prepared with the same parameters as
154 single layer coating (threaded rod with 1.27 diameter –100 μ m estimated wet thickness). Ag NWs only
155 coating was used as control and prior to deposition, the starting suspension was diluted to 0.5% wt in DI
156 and dispersed using an Ultra Turrax high shear disperser (30s, 10,000 rpm). The produced samples were
157 cut into 5 cm x 5 cm specimens and dry-sterilized for 24 hours at 60°C. 10 ml of pre-inoculated PCA at

158 10^5 CFU.ml⁻¹ (Colony Forming Unit) was poured in a petri dish and after cooling the samples were put on
159 top with the coated side facing the agar. The incubation period was 72 hours at 37°C and the inhibition
160 zone was assessed visually after incubation.

161 Quantitative tests were performed on single layer coatings with different thickness using a protocol
162 adapted from the AATCC TM100-1998 standard (AATCC standard, 1998). The samples were cut into 2
163 cm x 2 cm specimens and dry-sterilized for 24 hours at 60°C. 200 µl of bacterial suspension at 5.10^5
164 CFU.mL⁻¹ in 20% nutrient broth (5 g.l⁻¹ nutrient broth, 6.8 g.l⁻¹ sodium chloride) were deposited on the
165 samples specimen and incubated at 37°C for 24 hours. After incubation, the bacterial suspension was
166 recovered by washing with a prepared neutralizing solution (3 g.l⁻¹ of L- α -phosphatidyl choline, 5 g.l⁻¹ of
167 sodium thiosulphate, 1 g.l⁻¹ of L-histidine, 30 g.l⁻¹ of Tween 80, 10 ml of potassium dihydrogen phosphate
168 at 0.0425 g.l⁻¹, controlled pH of 7.2 ± 0.2) and the final concentration was determined by the plate
169 counting numbering method. The antibacterial activity (AA, %) of the samples was compared
170 quantitatively using the following equation (1):

$$171 \quad AA (\%) = \frac{BC_{reference} - BC_{sample}}{BC_{reference}} \times 100 \quad (1)$$

172 Where BC_{reference} (log CFU) is the remaining bacterial concentration on the PET substrate with no coating
173 and BC_{sample} (log CFU) is the remaining concentration on the tested sample.

174 **2.5. Barrier properties characterization**

175 The Oxygen Transmission Rate (OTR) of the coated samples was measured at 0, 50 and 80 % humidity
176 rate following the ASTM-F 1927-98 standards (ASTM standard, 2014) on a Systech Illinois Permeation
177 Analyser (USA) equipped with a coulometric detector. The tests were conducted at 23°C and with a 6.15
178 cm⁻¹ exchange surface. At least 3 different samples for each coating were tested and an average is
179 computed along with the associated standard deviation.

180 Water Vapour Transmission Rate (WVTR) was also measured on the same samples following a slightly
181 modified T 448 om-09 standard (TAPPI Standard, 2009). Around 30 g of anhydrous calcium dichloride
182 salt was put in a metal cup covered by the test sample and closed by a rubber gasket and screw-down cap.
183 The test was conducted at a regulated temperature (23°C) and humidity (50 %). The water-uptake of the
184 anhydrous salt through the tested sample was monitored by weighing at least twice a day for at least one
185 week. The WVTR value was calculated using the following equation (2):

$$186 \quad WVTR (g.m^{-2}.day^{-1}) = \frac{24 \times \Delta m}{\Delta t \times S} \quad (2)$$

187 Where Δt (h) is the time between two measurements, Δm (g) the corresponding weigh-uptake and S (m²)
188 is the exposed surface (6.15 cm²).

189 **2.6. Specific ink formulation and characterization for up-scaling test**

190 A specific, lower cost and in bigger volume formulation was prepared using T-CNF suspension
191 (Betulium) and Ag NWs suspension (NanoGap) redispersed together at 1% wt total mass with a mass ratio
192 of 1:1 using an Ultra Turrax high shear disperser (30s, 10,000 rpm).

193 T-CNF was imaged using a Philips CM 200/FEI (USA) Transmission Electron Microscope (TEM)
194 equipped with a TemCam F216 from TVIPS (Germany), at 200 kV acceleration voltage. A diluted
195 suspension of T-CNF was drop-casted on a copper grid with an amorphous carbon coating and uranyl
196 acetate (2%) dye was deposited on the drop-casted sample. Ag NWs were imaged by Atomic Force
197 Microscopy (AFM) and Scanning Electron Microscopy (SEM). AFM images were recorded on a Veeco
198 NanoScope-V apparatus (Canada). Ag NWs suspension was drop-casted on a Mica substrate and dried
199 overnight. A tapping mode with an OTESPA Bruker (USA) silicon cantilever was used covering a 3.3 μm
200 x 3.3 μm surface area.

201 The sedimentation evolution of the formulated ink was characterized by putting 10 ml of ink into a
202 cylindrical glass vial of 29 mm diameter section and measuring the sediment height after a fixed time. The
203 sediment height (S , %) was calculated using the following equation (3):

$$204 \quad S = \frac{H}{H_0} \times 100 \quad (3)$$

205 Where H (mm) is the measured sediment height and H_0 (mm) the total liquid height.

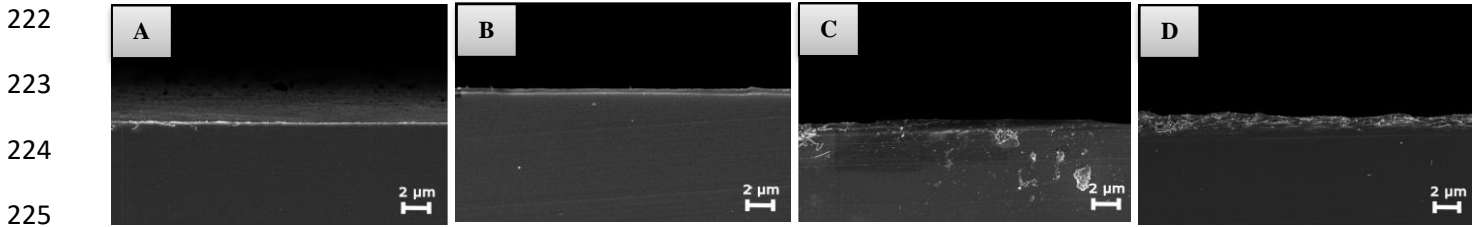
206 The rheological studies were performed on a rotational rheometer (MCR02, Anton Paar) at a set
207 temperature of 20°C with a cone-plate geometry (1.013° angle, 50 mm diameter and 55 μm truncate)
208 along with a set gap of 0.112 mm. Flow curves experiment were performed with increasing shear rate
209 ranging from 0.1 to 1000 s⁻¹.

210 Reverse gravure coating was conducted using a Mini-LaboTM apparatus from MIRWEC Film, Inc./ Yasui
211 Seiki company (USA). Coating was applied at a speed of 1 m.min⁻¹ and dried using infrared heaters. Two
212 different coating weights were produced using two different gravure rolls: roll 30 (estimated surface
213 volume of 150 cm³.m⁻², estimated transfer fraction of 0.33 and estimated approximate wet coat thickness
214 of 30-45 μm), and roll 120 (estimated surface volume of 34.7 cm³.m⁻², estimated transfer fraction of 0.28
215 and estimated approximate wet coat thickness of 5-11 μm).

216 **3. Results and discussion**

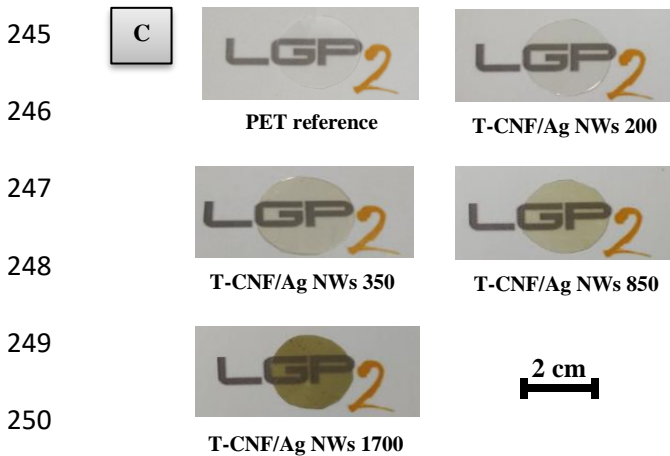
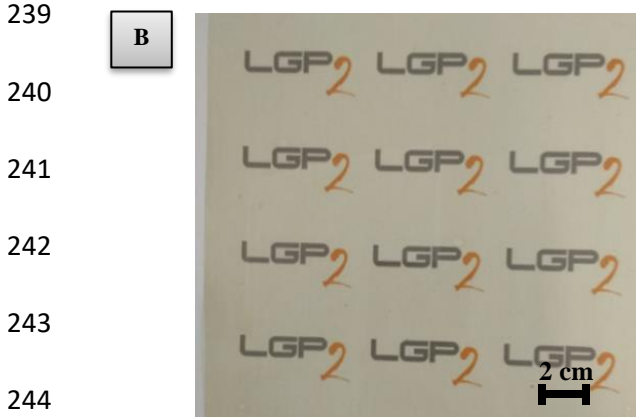
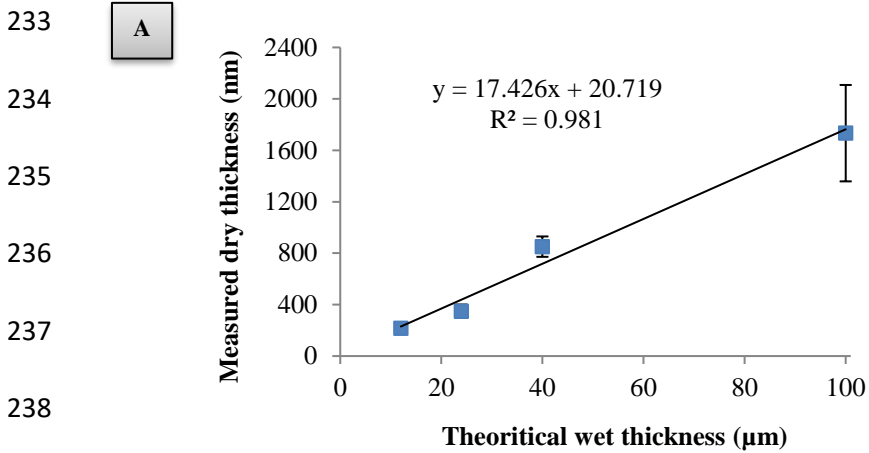
217 **3.1. Bar coating deposition on PET and antibacterial properties.**

218 The active ink was bar coated at 4 different thicknesses on PET sheets and the dry thickness was
219 determined from a cross section captured by FEG-SEM imaging (Figure 1). For the thicker coatings the
220 results demonstrated a higher standard deviation because the ultramicrotome cuts damaged the coated
221 layer.



226 **Figure 1: FEG-SEM cross section images of A) 12 μm wet estimated thickness coating B) 24 μm wet**
227 **estimated thickness coating C) 40 μm wet estimated thickness coating D) 100 μm wet estimate thickness**
228 **coating**

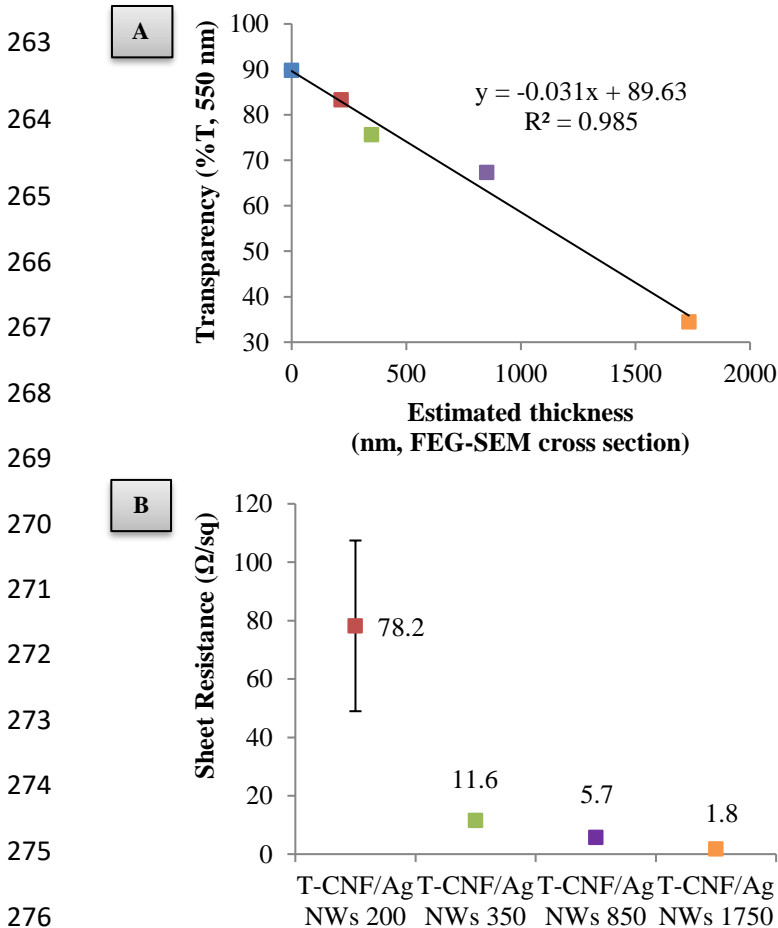
228 The measured average thicknesses are respectively 217 ± 27 , 349 ± 52 , 851 ± 79 and 1733 ± 375 nm. A
229 good linear correlation was found between wet and dry values (Figure 2**Error! Reference source not**
230 **found.**A) proving a controlled thickness deposition and accurate measurement. Consequently, in this
231 study the samples will be designated T-CNF/Ag NWs 200, T-CNF/Ag NWs 350, T-CNF/Ag NWs 850,
232 T-CNF/Ag NWs 1750, referred according to their dry thickness.



251 **Figure 2: Coating quality investigated by A) measured**
 252 **dried thickness (nm) vs theoretical wet thickness (µm), B)**
 253 **qualitative picture of 15 cm x 15 cm large area of T-**
 254 **CNF/Ag NWs 350 sample and B) qualitative picture of**
 255 **reduced area of all produced samples (2 cm diameter) and**
 256 **scale bar.**

254 From a macroscopic point of view, the coatings were observed to be uniform with no local uncoated
 255 patches (Figure 2B). For packaging applications, one of the key parameters for active coatings is to retain

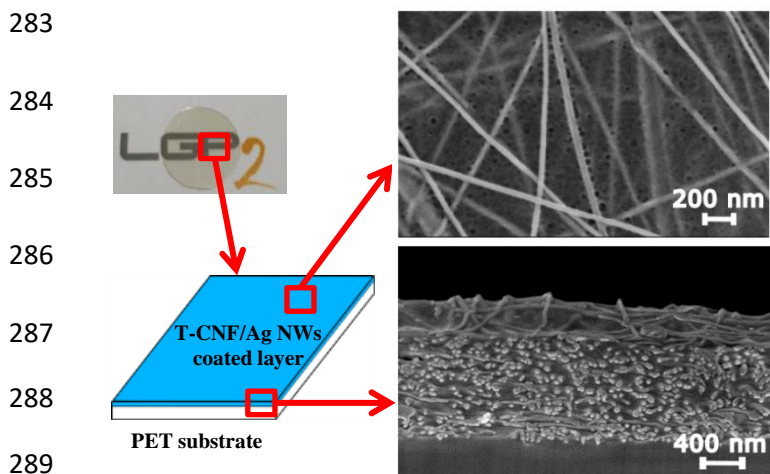
256 transparency for the purpose of aesthetics and product display. The coatings were found to be relatively
 257 transparent as the LGP2 logo is clearly visible under the different samples displayed in Figure 2C. The
 258 transmittance at 550 nm was measured at 83.3, 75.6 and 67.3% respectively for T-CNF/Ag NWs 200, T-
 259 CNF/Ag NWs 350 and T-CNF/Ag NWs 850 samples, when compared to the 89.8% reference uncoated
 260 PET. Coating quality may also be determined by measuring conductivity (Fahad et al., 2019) and except
 261 for T-CNF/Ag NWs 200, the produced coatings display high conductivity (low sheet resistance) with a
 262 very low standard deviation, confirming coating uniformity (Figure 3B).



277 **Figure 3: Coatings quality assessment using A)**
 278 **transparency measurement (%T, 550 nm) vs**
 279 **estimated dry thickness (nm) and B) sheet resistance**
 280 **measurements by 4-probe system (Ω/sq).**

279 The morphology of the coated layer was also investigated using electron microscopy imaging (Figure 4)
 280 and the coated layer was found to be highly organized in a dense network, yet not aggregated, with an
 281 orientation of the Ag NWs toward the direction of shearing during the bar coating process.

282

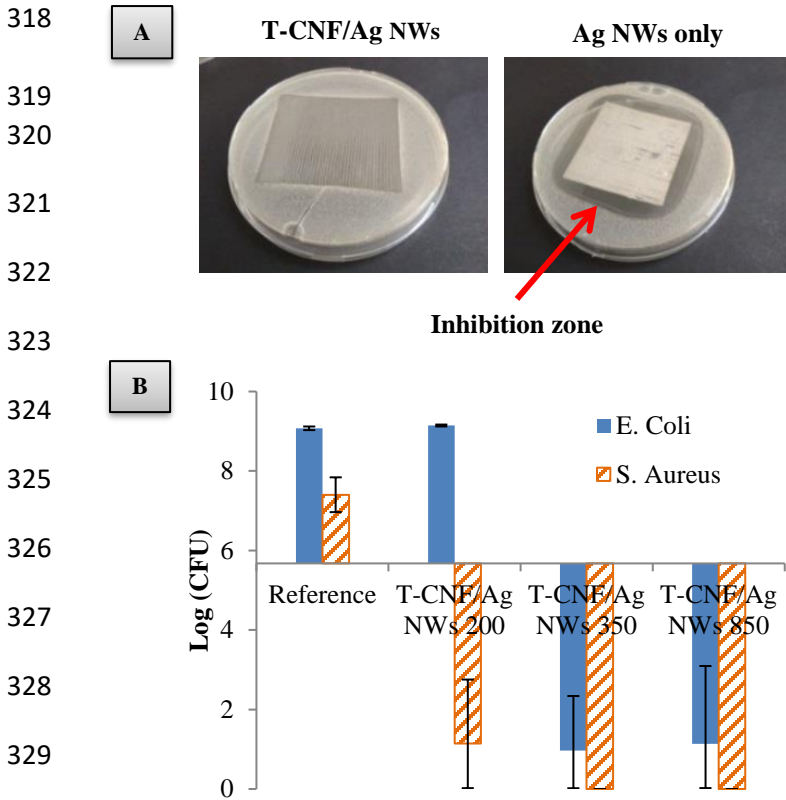


290 **Figure 4: Schematic view of the T-CNF/Ag NWs 850**
 291 **coated samples and FEG-SEM pictures of surface and**
 292 **cross-section**

292 A preliminary leaching study was performed prior to antibacterial characterization. The test showed that
 293 no peaks corresponding to Ag NWs could be seen after recording UV-vis spectra of the recovered liquid
 294 media after the leaching assay (Supporting Information). It can then be concluded that there is only limited
 295 leaching of Ag NWs measured by this technique. The antibacterial activity of the active coatings was then
 296 investigated with qualitative inhibition zone test (Figure 5A). For the sample coated only with Ag NWs, a
 297 broad inhibition zone is visible proving the antibacterial activity of Ag NWs. The antibacterial activity of
 298 Ag NWs was established to be mainly due to the release of silver ions because of their lower surface area
 299 and thus lower adhesion with bacteria cell walls compared to spherical or cubic silver nanoparticles (Hong
 300 et al., 2016; Visnapuu et al., 2013). It can then be assumed that the broad inhibition zone of the sample is
 301 due to the release of silver ions into the media. For the sample coated with the mixture of both T-CNF and
 302 Ag NWs no inhibition zone can be seen which suggests that physically entrapping the Ag NWs within the
 303 T-CNF matrix leads to a contact killing mode of action with no or minimal release of active material.

304 To quantitatively verify the activity of the sample the AATCC standard method was performed against
 305 both Gram-positive *Staphylococcus Aureus* and Gram-negative *Escherichia Coli* (Figure 5B). The T-
 306 CNF/Ag NWs 200 sample show a surprising activity: it displays no impact against *E. Coli* whereas for *S.*
 307 *Aureus*, a strong reduction of 6.3 log of colony forming unit compared to the uncoated reference sample
 308 was noted, which corresponds to a 86.5% activity. Indeed, the bacterial activity in the presence of silver
 309 nanoparticles is usually higher for *E. Coli* than *S. Aureus* (Chernousova & Epple, 2013). This is explained
 310 by the difference in the bacteria cell wall: Gram-positive bacteria display a more permeable layer made of
 311 peptidoglycan only whereas Gram-negative bacteria have a thinner yet more impermeable
 312 lipopolysaccharide layer coupled to a thin peptidoglycan layer (Slavin et al., 2017). The difference in

313 activity observed when compared with the literature could be explained by the very thin coating layer for
 314 T-CNF/Ag NWs 200 which is probably close to the antibacterial activity limit of the coating. The T-
 315 CNF/Ag NWs 350 and T-CNF/Ag NWs 850 coatings displayed more or less the same performance
 316 respectively 89.3% vs *E. Coli*/100% vs *S. Aureus* and 87.6% vs *E. Coli*/100% vs *S. Aureus*, showing that a
 317 350 nm thick coating is enough to achieve almost 100% antibacterial activity.

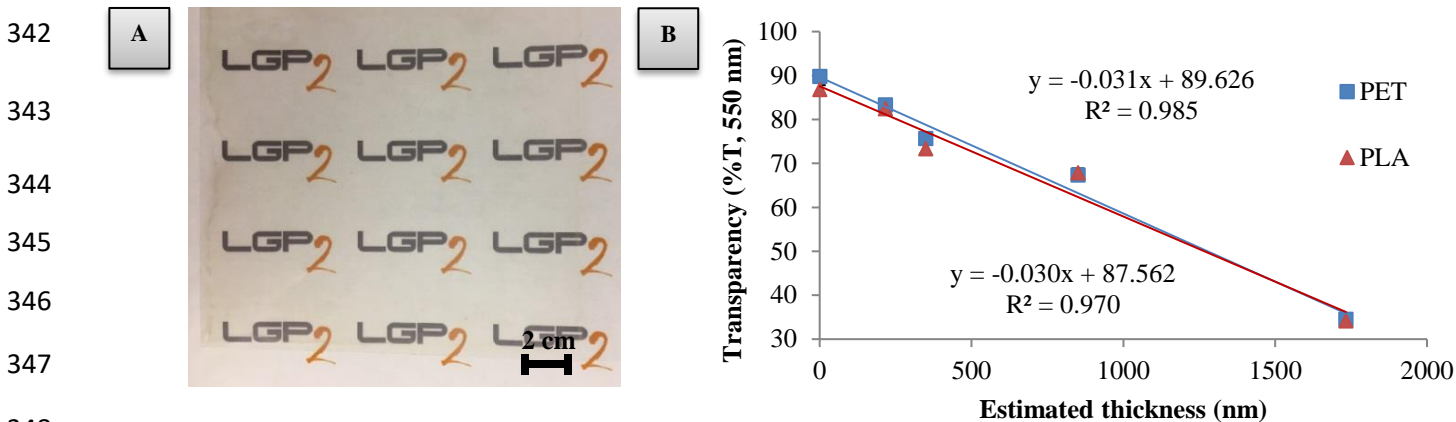


330 **Figure 5: Antibacterial characterization of the coated samples by A) inhibition qualitative analysis for *E. Coli* bacteria strain B) qualitative analysis using AATCC standard method 100-1998 vs *E. Coli* and *S. Aureus* with results expressed in log of colony forming units (log CFU).**

334 3.2. Bar coating on PLA and barrier properties.

335 In order to explore a bio-based substrate, the same ink was also deposited on polylactic acid (PLA) films
 336 using the bar coating process. The coatings were found to be homogeneous, with no uncoated patches and
 337 no visual defects (Figure 6A). The different deposited thickness for the PLA coatings were estimated to be
 338 relatively close to the ones for the PET coatings. Indeed the decrease of the transparency of the reference
 339 films due to the coating are more or less the same for both the PET and PLA coatings (Figure 6B). The

340 samples were then named with the same code, for instance T-CNF/Ag NWs 350 PLA corresponds to the
 341 T-CNF/Ag NWs coating on PLA with an approximated thickness of 350 nm.



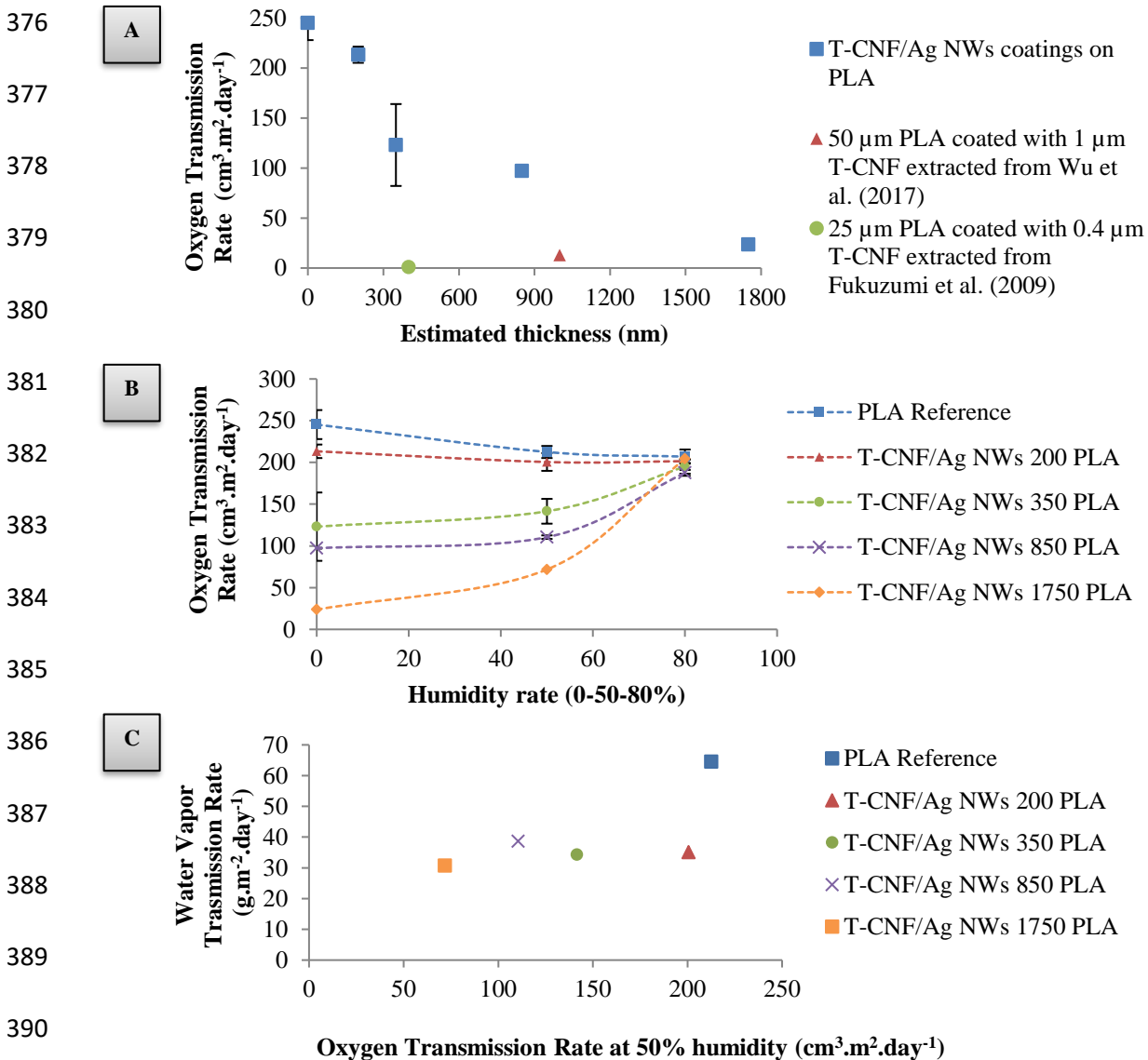
349 **Figure 6: T-CNF/Ag NWs coating on PLA quality assessment with A) picture of 15 cm x 15 cm large area of T-CNF/Ag NWs 350 sample and B) transparency (%T, 550nm) measurements and comparison with coatings on PET.**

350 CNF are often used in packaging application for their interesting barrier properties (Ferrer et al., 2017)
 351 and so the barrier properties of the coated PLA was investigated. Under dry conditions, the oxygen
 352 transmission rate (OTR) of the produced films was significantly decreased achieving more than 90%
 353 reduction for an estimated 1750 nm thick coatings (Figure 7A) but the measured values are however
 354 higher than those classically found in the literature for a T-CNF coating (Fukuzumi et al., 2009; Wu et al.,
 355 2017) (Figure 7A). Surprisingly in this study coating only T-CNF material was not possible due to high
 356 dewetting defect and so it was not possible to compare T-CNF/Ag NWs coatings to T-CNF coating only.
 357 A stronger surface treatment may be required to achieve T-CNF only coatings. CNF performance as a
 358 barrier layer is generally explained first by the large specific surface area and dense network of hydrogen
 359 bonding of the material which makes it difficult for any molecules to pass through (Ferrer et al., 2017).
 360 Then, CNF network also presents high tortuosity due to the impermeable crystalline regions and strong
 361 entanglement of the flexible fibres presenting a capability to seal any gaps within the network (Lagaron et
 362 al., 2004; Syverud & Stenius, 2008; Belbekhouche et al., 2011). Adding larger and rigid Ag NWs into the
 363 T-CNF matrix probably lead to a physical disruption of the sealed network and thus explaining the
 364 difference between the measured properties and the ones described in the literature.

365 As expected from using CNF materials, the OTR measurement is humidity dependent (Figure 7B) and at
 366 high humidity (80%) the OTR values of the different coatings are close to the uncoated PLA reference.
 367 The different coatings also showed a significant improvement of the WVTR values (Figure 7C) which is
 368 relatively similar (around 50% reduction) for all of the samples. This is interesting as coating CNF on
 369 polymer substrate usually leads to unchanged or slightly reduced water vapor permeability (Aulin et al.,

370 2013; Vartiainen et al., 2017), due to the hydrophilic nature and low water resistance of such materials.
 371 (Nair et al., 2014).

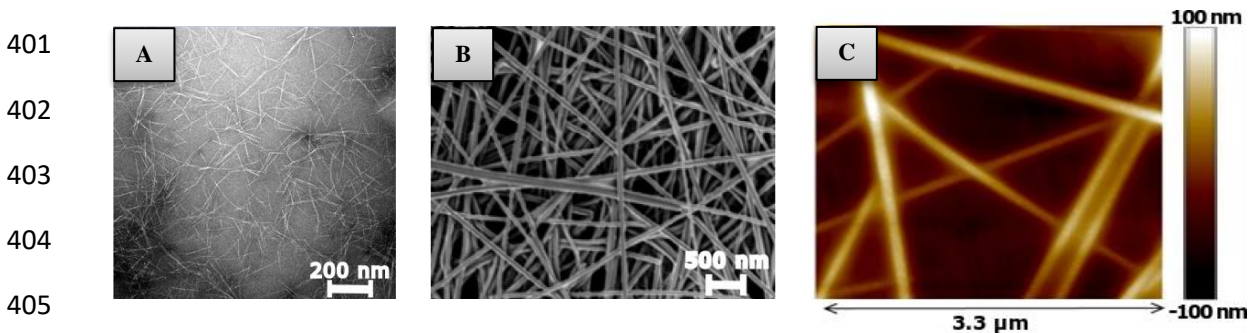
372 Coatings on biopolymer PLA sheets were successfully conducted and the coatings were found to be
 373 homogeneous with no uncoated patches and no visual defects. The deposited layers were found to be
 374 relatively similar to the coatings on PET and significant decrease for both the oxygen transmission rate
 375 and water vapour transmission rate was noted, proving the enhanced barrier properties.



391 **Figure 7: PLA coatings barrier properties characterizations with A) Oxygen Transmission Rate**
 392 **($\text{cm}^3 \cdot \text{m}^{-2} \cdot \text{day}^{-1}$) measurements in dry conditions vs estimated coating thickness (nm) and**
 393 **comparison with literature data B) Oxygen Transmission Rate ($\text{cm}^3 \cdot \text{m}^{-2} \cdot \text{day}^{-1}$) measurements at**
different humidity rate (0-50-80%) and C) Water Vapor Transmission Rate ($\text{g} \cdot \text{m}^{-2} \cdot \text{day}^{-1}$) vs
Oxygen Transmission Rate ($\text{cm}^3 \cdot \text{m}^{-2} \cdot \text{day}^{-1}$) at 50% of humidity.

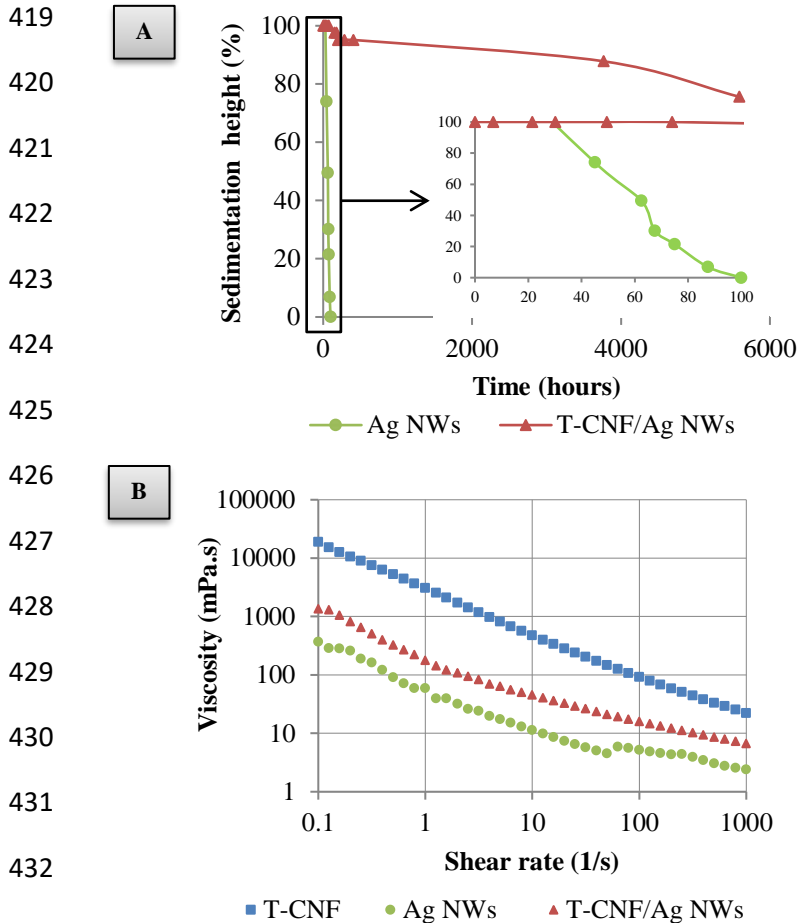
394 3.3. Up-scaling

395 As a more direct route and from a more economical point of view, a similar antibacterial ink was
396 specifically formulated by simple mixing and redispersion of the separately supplied components T-CNF
397 and Ag NWs in deionized water (see Materials and methods). The purchased T-CNF possess small
398 dimensions (approximately length 50-400 nm, diameter: 5-15 nm) and are rather rigid, as shown in the
399 TEM images in Figure 8A. The silver nanowires dimensions were approximately one order of magnitude
400 higher than the T-CNF: estimated length 10-50 μm , diameter 50-100 nm (Figure 8B and C).



406 **Figure 8: Raw materials microscopy imaging for A) TEM images of T-CNF B) SEM**
407 **images of silver nanowires C) AFM images of silver nanowires**

408 As it has already been described elsewhere (Hoeng et al., 2016), the T-CNF/Ag NWs ink is relatively
409 stable over time and exhibits a very low sedimentation rate, for example 75% sedimentation height after 8
410 months of testing, as compared to the Ag NWs suspension without T-CNF, that sedimented completely
411 after 4 days (Figure 9A). Flow curve rheology experiments were performed to assess the possibility to use
412 coating process for ink deposition and both materials and the final formulation display classic strong shear
413 thinning behaviour (Herrick et al., 1983; Hoeng et al., 2017) (Figure 9B). These behaviours can be
414 explained by the high aspect ratio of the nanoparticles in these colloidal suspensions and the induced
415 alignment of the particles at high shear rate, thus decreasing the viscosity. Ink viscosity at high shear rate
416 was found to be in line with was expected for the bar coating process. It was found that for the different
417 tested shear rate, viscosity of T-CNF suspension was one order of magnitude higher than Ag NWs. T-CNF
418 dictates the viscosity in the T-CNF/Ag NWs ink by acting as a rheological modifier which was already
shown in the literature for other aqueous CNF suspension coatings (Grüneberger et al., 2014).



434 **Figure 9: Ink formulation characterization compared to raw materials with A) sedimentation height measurement (%) and B) rheological flow curves. All formulations are at 1% wt total concentration.**

436 **3.3. Roll-to-roll process deposition**

437 The potential for this formulation to be applied to industrial-like roll-to-roll coating process was also

438 investigated using a reverse gravure coating process. Reverse gravure coating is interesting for packaging

439 applications because of the wide range of inks and substrate that can be used and was also chosen because

440 of the versatility of the process and the possibility to deposit low coat weights with a uniform and

441 controlled thickness (Kipphan, 2001; Vak et al., 2016). Two different coating thicknesses were

442 investigated and designated after their estimated wet thickness and previously established wet

443 thickness/dry thickness relation for the bar-coated samples. In other words, T-CNF/Ag NWs 850RR for

444 the coating using the roll 30 (wet coat thickness of 30-45 μm), and T-CNF/Ag NWs 200RR for the

445 coating using roll 120 (wet coat thickness of 5-11 μm). The coatings were found to be visually of high

446 quality with no uncoated patches, no visual defects and good homogeneity (Figure 10A). The antibacterial

447 properties of the T-CNF/Ag NWs 200RR is relatively low whereas T-CNF/Ag NWs 850RR display a

448 strong antimicrobial effect (Figure 10B). The T-CNF/Ag NWs 850RR sample showed a strong and
 449 significant bactericidal effect, precisely 7.22 log reduction corresponding to 77.9% calculated antibacterial
 450 activity against *E. Coli* and 100% against *S. Aureus*. On the other hand, the T-CNF/Ag NWs 200RR
 451 sample showed no activity at all against *S. Aureus* whereas it showed a 7.57 log reduction corresponding
 452 to 81.7% activity against *E. Coli*. The measured antibacterial activity for the roll-to-roll coated samples
 453 using the up-scaled ink formulation were found to be in line with the results for the bar coated samples
 454 and thus the proposed solution for antibacterial packaging could hence be easily adapted to an industrial
 455 scale.

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A



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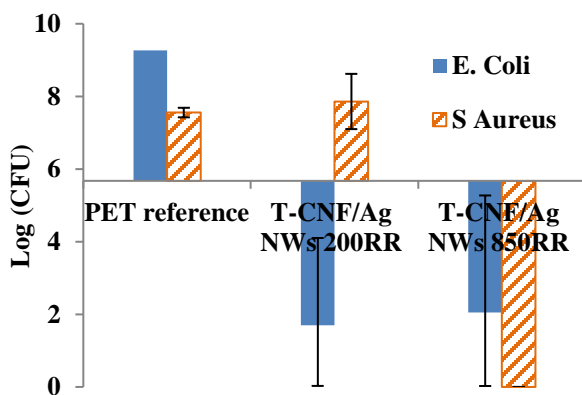
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B



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470 **Figure 10: Roll-to-roll coatings with A) visual aspect of the coatings and B) AATCC qualitative method for antibacterial activity assessment.**

471 **4. Conclusion**

472 An ink composed of silver nanowires and TEMPO-oxidised cellulose nanofibrils was successfully coated
473 first on flexible and transparent polyethylene terephthalate sheets. The controlled thickness of the coatings
474 ranged from 200 to 1750 nm with a high relative transparency superior to 65%, and the deposited layers
475 were found to be homogeneous with no visual defects. The antibacterial activity was suggested to work by
476 contact active killing mode of action and for the 350 nm thick coating, a significant AATCC standard
477 antibacterial activity of 89.3% and 100% was noted respectively against Gram-negative *Escherichia Coli*
478 and Gram-positive *Staphylococcus Aureus* bacteria. Moreover, the same ink was also deposited on PLA
479 sheets with similar estimated thickness and quality. After coating, oxygen and water vapour permeability
480 of the PLA substrate was significantly reduced, respectively 49% of oxygen transmission rate (dry
481 conditions) and 47% reduction of water vapor transmission rate for 350 nm thick coating. Finally, a
482 similar ink with lower cost materials was specifically formulated for up-scaling purposes and producing
483 active surfaces using a roll-to-roll process was proved using the reverse gravure process, showing a strong
484 retained antibacterial activity. Perspectives for a future work would be investigation of the migration of
485 silver ions within the cellulose nanofibrils matrix and the influence of matrix surface chemistry and the
486 investigation of the role of Ag NWs in the barrier properties of the system.

487 **Declaration of competing interest**

488 The authors declare that there is no conflict of interests with this manuscript.

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495 **References**

- 496 AATCC standard. (1998). *TM100 - Test Method for Antibacterial Finishes on Textile Materials:*
497 *Assessment of Antibacterial Textile*. AATCC, Research Triangle Park, NC.
- 498 AFNOR standard. (2005). *NF EN 1104 - Papier et carton destinés à entrer en contact avec les denrées*
499 *alimentaires : Détermination du transfert des constituants antimicrobiens*. AFNOR Association,
500 La Plaine Saint-Denis, France.
- 501 Afra, E., Mohammadnejad, S., & Saraeyan, A. (2016). Cellulose nanofibils as coating material and its
502 effects on paper properties. *Progress in Organic Coatings*, *101*, 455–460.
503 <https://doi.org/10.1016/J.PORGCOAT.2016.09.018>
- 504 ASTM standard. (2014). *F1927, 14 - Test Method for Determination of Oxygen Gas Transmission Rate,*
505 *Permeability and Permeance at Controlled Relative Humidity Through Barrier Materials Using a*
506 *Coulometric Detector*. ASTM International, West Conshohocken PA.
- 507 Aulin, C., Karabulut, E., Tran, A., Wågberg, L., & Lindström, T. (2013). Transparent Nanocellulosic
508 Multilayer Thin Films on Polylactic Acid with Tunable Gas Barrier Properties. *ACS Applied*
509 *Materials & Interfaces*, *5*(15), 7352–7359. <https://doi.org/10.1021/am401700n>
- 510 Belbekhouche, S., Bras, J., Siqueira, G., Chappey, C., Lebrun, L., Khelifi, B., Marais, S., & Dufresne, A.
511 (2011). Water sorption behavior and gas barrier properties of cellulose whiskers and microfibrils
512 films. *Carbohydrate Polymers*, *83*(4), 1740–1748. <https://doi.org/10.1016/j.carbpol.2010.10.036>
- 513 Carlson, C., Hussain, S. M., Schrand, A. M., Braydich-Stolle, L. K., Hess, K. L., Jones, R. L., & Schlager,
514 J. J. (2008). Unique cellular interaction of silver nanoparticles: size-dependent generation of
515 reactive oxygen species. *The Journal of Physical Chemistry. B*, *112*(43), 13608–13619.
516 <https://doi.org/10.1021/jp712087m>
- 517 Chernousova, S., & Epple, M. (2013). Silver as Antibacterial Agent: Ion, Nanoparticle, and Metal.
518 *Angewandte Chemie International Edition*, *52*(6), 1636–1653.
519 <https://doi.org/10.1002/anie.201205923>
- 520 El Badawy, A. M., Silva, R. G., Morris, B., Scheckel, K. G., Suidan, M. T., & Tolaymat, T. M. (2011).
521 Surface Charge-Dependent Toxicity of Silver Nanoparticles. *Environmental Science &*
522 *Technology*, *45*(1), 283–287. <https://doi.org/10.1021/es1034188>
- 523 Fahad, S., Yu, H., Wang, L., Zain-ul-Abdin, Haroon, M., Ullah, R. S., Nazir, A., Naveed, K.-R.,
524 Elshaarani, T., & Khan, A. (2019). Recent progress in the synthesis of silver nanowires and their
525 role as conducting materials. *Journal of Materials Science*, *54*(2), 997–1035.
526 <https://doi.org/10.1007/s10853-018-2994-9>
- 527 Ferrer, A., Pal, L., & Hubbe, M. (2017). Nanocellulose in packaging: Advances in barrier layer
528 technologies. *Industrial Crops and Products*, *95*(Supplement C), 574–582.
529 <https://doi.org/10.1016/j.indcrop.2016.11.012>
- 530 Food and Agriculture Organization of the United Nations. (2019). *World food and agriculture: statistical*
531 *pocketbook 2019*.
- 532 Fukuzumi, H., Saito, T., Iwata, T., Kumamoto, Y., & Isogai, A. (2009). Transparent and High Gas Barrier
533 Films of Cellulose Nanofibers Prepared by TEMPO-Mediated Oxidation. *Biomacromolecules*,
534 *10*(1), 162–165. <https://doi.org/10.1021/bm801065u>
- 535 Grüneberger, F., Künniger, T., Zimmermann, T., & Arnold, M. (2014). Rheology of nanofibrillated
536 cellulose/acrylate systems for coating applications. *Cellulose*, *21*(3), 1313–1326.
537 <https://doi.org/10.1007/s10570-014-0248-9>

538 Herrick, F. W., Casebier, R. L., Hamilton, J. K., & Sandberg, K. R. (1983). Microfibrillated Cellulose:
539 Morphology and Accessibility. *J. Appl. Polym. Sci.: Appl. Polym. Symp.; (United States)*, 37.
540 <https://www.osti.gov/scitech/biblio/5039044>

541 Hoeng, F., Denneulin, A., Krosnicki, G., & Bras, J. (2016). Positive impact of cellulose nanofibrils on
542 silver nanowire coatings for transparent conductive films. *Journal of Materials Chemistry C*,
543 4(46), 10945–10954. <https://doi.org/10.1039/C6TC03629E>

544 Hoeng, F., Denneulin, A., Reverdy-Bruas, N., Krosnicki, G., & Bras, J. (2017). Rheology of cellulose
545 nanofibrils/silver nanowires suspension for the production of transparent and conductive
546 electrodes by screen printing. *Applied Surface Science*, 394(Supplement C), 160–168.
547 <https://doi.org/10.1016/j.apsusc.2016.10.073>

548 Hong, X., Wen, J., Xiong, X., & Hu, Y. (2016). Shape effect on the antibacterial activity of silver
549 nanoparticles synthesized via a microwave-assisted method. *Environmental Science and Pollution
550 Research*, 23(5), 4489–4497. <https://doi.org/10.1007/s11356-015-5668-z>

551 Ilić, V., Šaponjić, Z., Vodnik, V., Potkonjak, B., Jovančić, P., Nedeljković, J., & Radetić, M. (2009). The
552 influence of silver content on antimicrobial activity and color of cotton fabrics functionalized with
553 Ag nanoparticles. *Carbohydrate Polymers*, 78(3), 564–569.
554 <https://doi.org/10.1016/j.carbpol.2009.05.015>

555 Kaur, R., & Liu, S. (2016). Antibacterial surface design – Contact kill. *Progress in Surface Science*, 91(3),
556 136–153. <https://doi.org/10.1016/j.progsurf.2016.09.001>

557 Kipphan, H. (Ed.). (2001). *Handbook of Print Media: Technologies and Production Methods*. Springer-
558 Verlag. <https://doi.org/10.1007/978-3-540-29900-4>

559 Korani, M., Ghazizadeh, E., Korani, S., Hami, Z., & Mohammadi-Bardbori, A. (2015). Effects of silver
560 nanoparticles on human health. *European Journal of Nanomedicine*, 7(1), 51–62.
561 <https://doi.org/10.1515/ejnm-2014-0032>

562 Kvítek, L., Panáček, A., Soukupová, J., Kolář, M., Večeřová, R., Prucek, R., Holecová, M., & Zbořil, R.
563 (2008). Effect of Surfactants and Polymers on Stability and Antibacterial Activity of Silver
564 Nanoparticles (NPs). *The Journal of Physical Chemistry C*, 112(15), 5825–5834.
565 <https://doi.org/10.1021/jp711616v>

566 Lagaron, J. M., Catalá, R., & Gavara, R. (2004). Structural characteristics defining high barrier properties
567 in polymeric materials. *Materials Science and Technology*, 20(1), 1–7.
568 <https://doi.org/10.1179/026708304225010442>

569 Martins, N. C. T., Freire, C. S. R., Pinto, R. J. B., Fernandes, S. C. M., Neto, C. P., Silvestre, A. J. D.,
570 Causio, J., Baldi, G., Sadocco, P., & Trindade, T. (2012). Electrostatic assembly of Ag
571 nanoparticles onto microfibrillated cellulose for antibacterial paper products. *Cellulose*, 19(4),
572 1425–1436. <https://doi.org/10.1007/s10570-012-9713-5>

573 Morones, J. R., Elechiguerra, J. L., Camacho, A., Holt, K., Kouri, J. B., Ramírez, J. T., & Yacaman, M. J.
574 (2005). The bactericidal effect of silver nanoparticles. *Nanotechnology*, 16(10), 2346.
575 <https://doi.org/10.1088/0957-4484/16/10/059>

576 Nair, S. S., Zhu, J., Deng, Y., & Ragauskas, A. J. (2014). High performance green barriers based on
577 nanocellulose. *Sustainable Chemical Processes*, 2, 23. <https://doi.org/10.1186/s40508-014-0023-0>

578 Pal, S., Tak, Y. K., & Song, J. M. (2007). Does the Antibacterial Activity of Silver Nanoparticles Depend
579 on the Shape of the Nanoparticle? A Study of the Gram-Negative Bacterium *Escherichia coli*.
580 *Applied and Environmental Microbiology*, 73(6), 1712–1720.
581 <https://doi.org/10.1128/AEM.02218-06>

582 Rai, M. k., Deshmukh, S. d., Ingle, A. p., & Gade, A. k. (2012). Silver nanoparticles: the powerful
583 nanoweapon against multidrug-resistant bacteria. *Journal of Applied Microbiology*, *112*(5), 841–
584 852. <https://doi.org/10.1111/j.1365-2672.2012.05253.x>

585 Ramaraju, B., Imae, T., & Destaye, A. G. (2015). Ag nanoparticle-immobilized cellulose nanofibril films
586 for environmental conservation. *Applied Catalysis A: General*, *492*(Supplement C), 184–189.
587 <https://doi.org/10.1016/j.apcata.2014.12.045>

588 Rodionova, G., Saito, T., Lenes, M., Eriksen, Ø., Gregersen, Ø., Fukuzumi, H., & Isogai, A. (2012).
589 Mechanical and oxygen barrier properties of films prepared from fibrillated dispersions of
590 TEMPO-oxidized Norway spruce and Eucalyptus pulps. *Cellulose*, *19*(3), 705–711.
591 <https://doi.org/10.1007/s10570-012-9664-x>

592 Schindelin, J., Arganda-Carreras, I., Frise, E., Kaynig, V., Longair, M., Pietzsch, T., Preibisch, S.,
593 Rueden, C., Saalfeld, S., Schmid, B., Tinevez, J.-Y., White, D. J., Hartenstein, V., Eliceiri, K.,
594 Tomancak, P., & Cardona, A. (2012). Fiji: an open-source platform for biological-image analysis.
595 *Nature Methods*, *9*(7), 676–682. <https://doi.org/10.1038/nmeth.2019>

596 Schneider, C. A., Rasband, W. S., & Eliceiri, K. W. (2012). NIH Image to ImageJ: 25 years of image
597 analysis. *Nature Methods*, *9*(7), 671–675. <https://doi.org/10.1038/nmeth.2089>

598 Slavin, Y. N., Asnis, J., Häfeli, U. O., & Bach, H. (2017). Metal nanoparticles: understanding the
599 mechanisms behind antibacterial activity. *Journal of Nanobiotechnology*, *15*(1), 65.
600 <https://doi.org/10.1186/s12951-017-0308-z>

601 Smetana, A. B., Klabunde, K. J., Marchin, G. R., & Sorensen, C. M. (2008). Biocidal Activity of
602 Nanocrystalline Silver Powders and Particles. *Langmuir*, *24*(14), 7457–7464.
603 <https://doi.org/10.1021/la800091y>

604 Sofi, S. A., Singh, J., Rafiq, S., Ashraf, U., Dar, B. N., & Nayik, G. A. (2018). A Comprehensive Review
605 on Antimicrobial Packaging and its Use in Food Packaging. *Current Nutrition & Food Science*,
606 *14*(4), 305–312. <https://doi.org/10.2174/1573401313666170609095732>

607 Syverud, K., & Stenius, P. (2008). Strength and barrier properties of MFC films. *Cellulose*, *16*(1), 75.
608 <https://doi.org/10.1007/s10570-008-9244-2>

609 TAPPI Standard. (2009). *T 448 om-09 - Water vapor transmission rate of paper and paperboard at 23°C*
610 *and 50% RH*. Technical Association of the Pulp and Paper Industry, New York.

611 Turbak, A. F., Snyder, F. W., & Sandberg, K. R. (1983). Microfibrillated Cellulose, a New Cellulose
612 Product: Properties, Uses, and Commercial Potential. *J. Appl. Polym. Sci.: Appl. Polym. Symp.;*
613 *(United States)*, *37*. <https://www.osti.gov/scitech/biblio/5062478>

614 Uddin, K. M. A., Orelma, H., Mohammadi, P., Borghei, M., Laine, J., Linder, M., & Rojas, O. J. (2017).
615 Retention of lysozyme activity by physical immobilization in nanocellulose aerogels and
616 antibacterial effects. *Cellulose*, *24*(7), 2837–2848. <https://doi.org/10.1007/s10570-017-1311-0>

617 Vak, D., Weerasinghe, H., Ramamurthy, J., Subbiah, J., Brown, M., & Jones, D. J. (2016). Reverse
618 gravure coating for roll-to-roll production of organic photovoltaics. *Solar Energy Materials and*
619 *Solar Cells*, *149*, 154–161. <https://doi.org/10.1016/j.solmat.2016.01.015>

620 Vartiainen, J., Laine, C., Willberg-Keyriläinen, P., Pitkänen, M., & Ohra-aho, T. (2017). Biobased
621 mineral-oil barrier-coated food-packaging films. *Journal of Applied Polymer Science*, *134*(9).
622 <https://doi.org/10.1002/app.44586>

623 Visnapuu, M., Joost, U., Juganson, K., Künnis-Beres, K., Kahru, A., Kisand, V., & Ivask, A. (2013).
624 Dissolution of Silver Nanowires and Nanospheres Dictates Their Toxicity to Escherichia coli.
625 *BioMed Research International*, *2013*, 1–9. <https://doi.org/10.1155/2013/819252>

- 626 Wu, B., Geng, B., Chen, Y., Liu, H., Li, G., & Wu, Q. (2017). Preparation and characteristics of TEMPO-
627 oxidized cellulose nanofibrils from bamboo pulp and their oxygen-barrier application in PLA
628 films. *Frontiers of Chemical Science and Engineering*, *11*(4), 554–563.
629 <https://doi.org/10.1007/s11705-017-1673-8>
- 630 Xu, Y., Li, S., Yue, X., & Lu, W. (2017). Review of Silver Nanoparticles (AgNPs)-Cellulose Antibacterial
631 Composites. *BioResources*, *13*(1), 2150-2170–2170. <https://doi.org/10.15376/biores.13.1.Xu>
- 632 Yildirim, S., Röcker, B., Pettersen, M. K., Nilsen-Nygaard, J., Ayhan, Z., Rutkaite, R., Radusin, T.,
633 Suminska, P., Marcos, B., & Coma, V. (2018). Active Packaging Applications for Food.
634 *Comprehensive Reviews in Food Science and Food Safety*, *17*(1), 165–199.
635 <https://doi.org/10.1111/1541-4337.12322>
- 636 Yu, Z., Wang, W., Kong, F., Lin, M., & Mustapha, A. (2019). Cellulose nanofibril/silver nanoparticle
637 composite as an active food packaging system and its toxicity to human colon cells. *International*
638 *Journal of Biological Macromolecules*, *129*, 887–894.
639 <https://doi.org/10.1016/j.ijbiomac.2019.02.084>